

Microwave dielectric properties and sintering behaviors of scheelite compound CaMoO_4

Geun-Kyu Choi^a, Seo-Yong Cho^b, Jae-Sul An^a, Kug Sun Hong^{a,*}

^a School of Materials Science and Engineering, College of Engineering, Seoul National University, Seoul, South Korea

^b Cerelectron, 284 Kalgot, Jinwee, PyungTaek, Kyunggi, South Korea

Available online 8 November 2005

Abstract

The microwave dielectric properties, sintering behaviors of scheelite compound CaMoO_4 were investigated using dilatometry, X-ray diffraction, scanning electron microscopy and network analyzer. To improve the sintering property of CaMoO_4 , samples with different Ca/Mo ratio were prepared. The bulk densities of $\text{CaMo}_{(x)}\text{O}_4$ (for $x = 1.02, 1.05, 1.08$) samples were higher than those of pure CaMoO_4 over all temperature range. The well-sintered $\text{CaMo}_{(x)}\text{O}_4$ (for $x = 1.02$) sample with $\sim 95.7\%$ of the theoretical density shows $Q \times f$ value of 71,000 GHz and dielectric constant (ϵ_r) = 10.3. The effects of liquid phase formation on sintering process and dielectric properties were discussed. Sample with hot-press sintering was also prepared in order to investigate dielectric properties of fully dense CaMoO_4 , which exhibited quality factor ($Q \times f$), $\sim 55,000$ GHz and dielectric constant (ϵ_r) = 11.7.

© 2005 Elsevier Ltd. All rights reserved.

Keywords: Hot pressing; Sintering; Porosity; Substrates

1. Introduction

Recently, Molybdenum (Mo) containing mixed oxides have been of practical interest because of their attractive luminescence property¹ and possibility of negative electrode (anode) materials.^{2,3} The scheelite compound has been investigated as a host crystal for laser applications. Also, the dielectric constant and the anisotropy of the dielectric constant (ϵ') of single crystals (grown by the Czochralski method) were firstly reported by Brower and Fang.⁴ The dielectric constant (ϵ') parallel to a -axis of CaMoO_4 single crystal was 24.0 ± 0.2 ; parallel to c -axis, 20.0 ± 0.2 at 1.59 kHz. The dielectric loss ($\tan \delta$) for both orientations of CaMoO_4 specimen was 10^{-3} or less. Unfortunately, not many works have been done on the dielectric properties for microwave application.

Dielectric ceramic substrate with low permittivity and high- Q is required to minimize cross-coupling with conductors and to maintain overall high- Q circuits by lowering power dissipation. Alumina ($\epsilon_r = 10$, $Q \times f = 100,000$, $\tau_f = -50$ ppm/ $^\circ\text{C}$) is the standard material for the manufacturing of hybrid circuits via thick or thin film technology and for high integrated multilayer circuits

produced via co-firing of metallized green tapes. We thought that CaMoO_4 ceramic could be selected as a possible candidate for microwave dielectric material because of its low permittivity and high- Q ($\epsilon_r = 11.7$, $Q \times f = 55,000$, $\tau_f = -60$ ppm/ $^\circ\text{C}$), such alumina.

To investigate possibility of microwave application of CaMoO_4 ceramic, the sintering behavior and microwave dielectric properties of dense CaMoO_4 ceramic were examined. In our study, as a way of obtaining fully dense ceramic, the specimens were uniaxially pressed during sintering at 1100 $^\circ\text{C}$ for 1 h under pressure of 16 MPa (hot-press method). But hot-press method has many problems on the commercial uses, so the samples with different Ca/Mo ratio were prepared to control sintering behavior of CaMoO_4 .

2. Experimental procedure

The powders were prepared by conventional mixed oxide method. CaCO_3 , MoO_3 (High Purity Chemical Lab., Japan) powders with 99.99 and 99.9% purity were weighed and mixed for 24 h with stabilized zirconia media and ethanol. The mixed slurry was dried and then powders were calcined at 800–1100 $^\circ\text{C}$ for 2 h in air and then the calcined powder was granulated with an organic binder after ball milling for 24 h. The milled powders

* Corresponding author. Tel.: +82 2 880 8024; fax: +82 2 886 4156.
E-mail address: kshongss@plaza.snu.ac.kr (K.S. Hong).

were uniaxially pressed into disks 8 mm in diameter and 2–4 mm thick under a pressure of 1000 kg/cm². Pellets were sintered at 900–1350 °C for 2 h in air with a heating rate of 5 °C/min. The bulk density of the sintered specimens was determined by the Archimedes' method.

To obtain dense samples, the specimens were uniaxially pressed during sintering at 1100 °C for 1 h under pressure of 16 MPa. Shrinkage of the specimens during heating was measured using a horizontal-loading dilatometer with Al₂O₃ rams and boats (Model DII, 420C, Netzsch Instruments, Germany). The X-ray powder diffraction (XRD) studies were performed with X-ray diffractometer (XRD: Model M18XHF, Mac Science Instruments, Japan). Polished and thermally etched surfaces of sintered specimens were examined using scanning electron microscopy (SEM: Model XL20, Philips, The Netherlands). The microwave dielectric properties of sintered samples were measured at *x*-band frequencies (14–17 GHz) using a network analyzer (model HP8720C, Hewlett Packard, Palo Alto, CA). The quality factor ($Q \times f$) was measured by the transmission cavity method using a Cu cavity and Teflon supporter.⁵ Relative dielectric constant (ϵ_r) was measured using the post-resonator method⁶ and temperature coefficient of resonant frequency (τ_f) was measured using an Invar cavity in the temperature at range 20–85 °C.⁷

3. Results and discussion

3.1. Sintering behavior of CaMo_(x)O₄

Fig. 1 shows bulk densities of pure CaMoO₄ and the samples with different Ca/Mo ratio, which are CaMo_(x)O₄ (Ca/Mo = 1/*x* for *x* = 1.0, 1.02, 1.05, 1.08) as a function of sintering temperature. The densities of pure CaMoO₄ sample significantly decreased at temperature range from 1200 to 1300 °C and has the maximum value of ~3.85 g/cm³ (~90.6% of the theoretical density) at 1000 °C. This result suggests that pure CaMoO₄

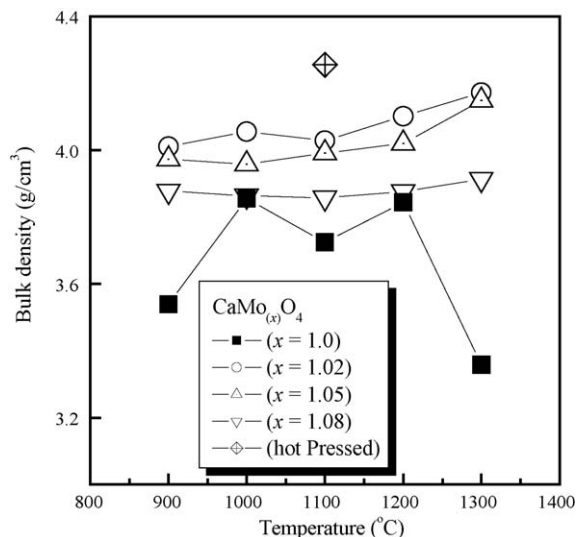


Fig. 1. Bulk densities of CaMo_(x)O₄ samples as a function of sintering temperature.

sample has poor densification characteristic. On the other hand, the densities of CaMo_(x)O₄ (for *x* = 1.02, 1.05, 1.08) samples slightly increased with increasing the sintering temperature and decreased with increasing *x*. The bulk density of CaMo_(x)O₄ (for *x* = 1.02) has the maximum value of ~4.07 g/cm³ (~95.7% of the theoretical density) at 1300 °C. The bulk density of hot pressed sample sintered at 1100 °C under pressure of 16 MPa was ~4.25 g/cm³ (~99.9% of the theoretical density). The bulk density of pure CaMoO₄ was lower than those of CaMo_(x)O₄ (for *x* = 1.02, 1.05, 1.08) over all temperature range. It was considered that densification characteristic of CaMo_(x)O₄ samples (for *x* = 1.02, 1.05, 1.08) was better than that of pure CaMoO₄ sample. The decrease in density of pure CaMoO₄ with increasing sintering temperature was revealed with microstructural investigation.

Fig. 2 shows SEM micrograph of the CaMo_(x)O₄ specimens (for *x* = 1.0, 1.02, 1.08) sintered at 1200–1300 °C for 2 h and hot-pressed at 1100 °C for 1 h under pressure of 16 MPa. The average grain size of pure CaMoO₄ samples increased abruptly from 4.5 (Fig. 2(a)) to 18 μm (Fig. 2(b)) with increasing sintering temperature from 1200 to 1300 °C. Dense microstructure was not achieved with increasing sintering temperature. Besides, the grains of CaMoO₄ sample sintered at 1300 °C for 2 h (Fig. 2(b)) have micro-crack. This result corresponds well the decrease in the density of pure CaMoO₄ with increasing sintering temperature from 1200 to 1300 °C. As shown in Fig. 2(c) and (d), the microscope images of hot pressed CaMoO₄ at 1100 °C for 1 h under 16 MPa and CaMo_(x)O₄ (for *x* = 1.02) show dense and fine microstructure. The average grain size of the CaMo_(x)O₄ sintered at 1200 °C increased from 3.5 (Fig. 2(d)) to 12 μm (Fig. 2(e)) with increasing *x* from 1.02 to 1.08. In addition of more excess Mo ions, grain growth was not accompanied with reduction of porosity, as shown in Fig. 2(e). The densification process is usually inhibited by rapid grain growth. This microstructure variation was consistent with the decrease in density of CaMo_(x)O₄ (for *x* = 1.02, 1.05, 1.08) with increasing *x*.

Fig. 3 shows shrinkage curves of the green compact of CaMo_(x)O₄ samples (for *x* = 1.0, 1.02, 1.05, 1.08). The shrinkage of pure CaMoO₄ occurred at broad temperature range (700–1200 °C) and final shrinkage (%) was less than 12%. The onset of CaMo_(x)O₄ (for *x* = 1.02, 1.05, 1.08) shrinkage occurred at lower temperature and the rapid shrinkage occurred at narrow temperature region (600–800 °C). As *x* increased, final shrinkage (%) was also increased. For CaMo_(x)O₄ (*x* = 1.08) sample, final shrinkage (%) was about ~15%. In CaO–MoO₃ phase diagram, MoO₃–CaMoO₄ system has a binary eutectic temperature at 726 ± 3 °C and MoO₃ = 25 wt.%.⁸ So sintered samples of CaMo_(x)O₄ (*x* = 1.02, 1.05, 1.08) could have solid CaMoO₄ phase and liquid MoO₃ phase over the eutectic temperature. Given that different Ca/Mo ratio system has more liquid MoO₃ phase with increasing *x*, which might suggest changes in final shrinkage (%). In Fig. 2(d) and (e), it is postulated that the abrupt grain growth of the CaMo_(x)O₄ sample sintered at 1200 °C with increasing *x* from 1.02 to 1.08 was caused by MoO₃ rich liquid phase. This results indicate that MoO₃ rich liquid phase could act as a sintering aid for CaMo_(x)O₄ (for *x* = 1.02, 1.05, 1.08).

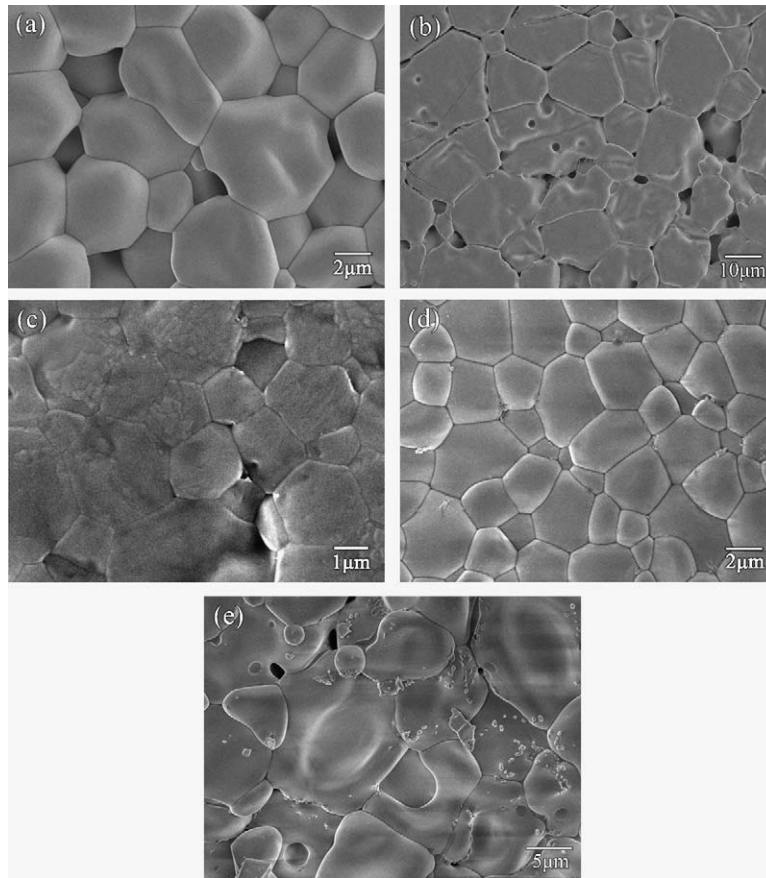


Fig. 2. SEM micrograph of the CaMoO_4 specimen: (a) sintered at $1200\text{ }^\circ\text{C}$ for 2 h, (b) sintered at $1300\text{ }^\circ\text{C}$ for 2 h, (c) hot-pressed at $1100\text{ }^\circ\text{C}$ for 1 h under pressure of 16 MPa, (d) $\text{CaMo}_{(x)}\text{O}_4$ ($x = 1.02$) sintered at $1200\text{ }^\circ\text{C}$ for 2 h, and (e) $\text{CaMo}_{(x)}\text{O}_4$ ($x = 1.08$) sintered at $1200\text{ }^\circ\text{C}$ for 2 h.

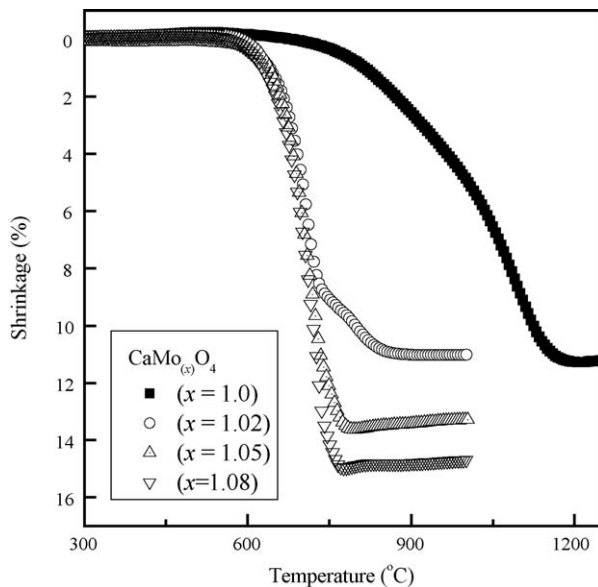


Fig. 3. Shrinkage curves of $\text{CaMo}_{(x)}\text{O}_4$ samples as a function of sintering temperature.

As shown in Fig. 4(a)–(d), with increasing x , remained MoO_3 phase was increased. It seemed that MoO_3 liquid phase would act as a sintering aid over the eutectic temperature and solely crystallized without formation of another phase under cooling.

It is considered that pure CaMoO_4 sample shows lower sinterability that is lower density and dielectric constant, as might be expected, is to be low with comparing that of CaMoO_4 (for $x = 1.02$). It is found that different Ca/Mo ratio could control the sintering behavior and improve the sintering properties of CaMoO_4 .

3.2. Microwave properties of $\text{CaMo}_{(x)}\text{O}_4$

Fig. 5 shows the microwave dielectric properties of $\text{CaMo}_{(x)}\text{O}_4$ samples sintered at temperatures range from 1000 to $1300\text{ }^\circ\text{C}$ for 2 h and CaMoO_4 sample sintered at $1100\text{ }^\circ\text{C}$ for 1 h under pressure of 16 MPa. In general, a higher density results in a higher dielectric constant owing to lower porosity ($\epsilon_r = 1$). Because the sinterability of pure CaMoO_4 was poor, pure CaMoO_4 sintered at $1100\text{ }^\circ\text{C}$ had lower dielectric constant (ϵ_r) of 9.2 than that of CaMoO_4 sample sintered at $1100\text{ }^\circ\text{C}$ under 16 MPa ($\epsilon_r = 11.7$), which well corresponds to density and SEM investigation.

In the case of different Ca/Mo ratio system, as x increased, dielectric constant of $\text{CaMo}_{(x)}\text{O}_4$ (for $x = 1.02, 1.05, 1.08$)

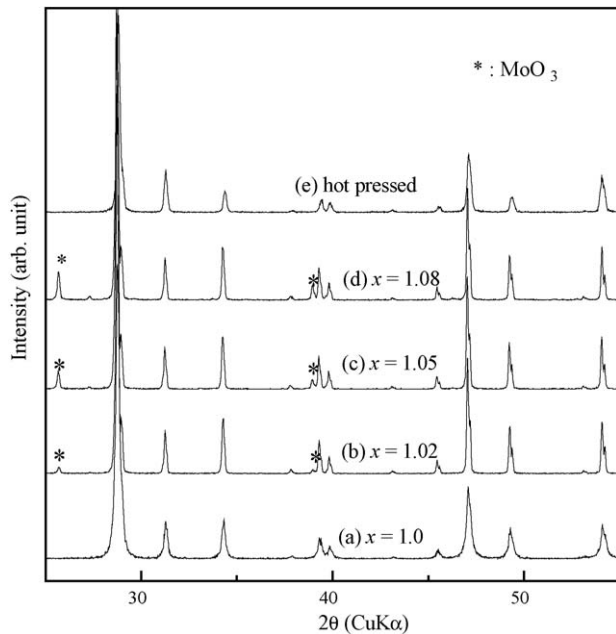


Fig. 4. XRD patterns of $\text{CaMo}_{(x)}\text{O}_4$ sintered at 1200°C for 2 h: (a) $x=1.0$, (b) $x=1.02$, (c) $x=1.05$, (d) $x=1.08$, and (e) hot pressed at 1100°C for 1 h under pressure of 16 MPa.

samples were decreased. In Section 3.1, we showed that grain growth with increasing x , which could be caused by MoO_3 rich liquid phase, could not reduce porosity in our system. It was found that the dielectric constant showed a close

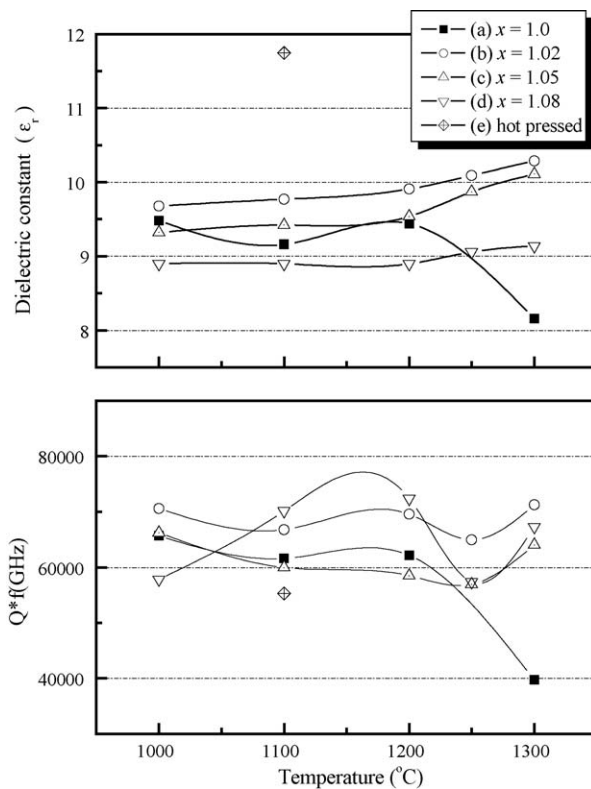


Fig. 5. Microwave properties of $\text{CaMo}_{(x)}\text{O}_4$ for: (a) $x=1.0$, (b) $x=1.02$, (c) $x=1.05$, (d) $x=1.08$, and (e) $x=1.0$ (hot pressed at 1100°C for 1 h under pressure of 16 MPa).

correlation with bulk density, as shown in Figs. 1 and 2. The presence of remaining MoO_3 ($\epsilon_r = 6.3 \pm 0.02$)⁹ phase could affect the decrease in the dielectric properties of $\text{CaMo}_{(x)}\text{O}_4$ samples, as shown in Fig. 4. For $\text{CaMo}_{(x)}\text{O}_4$ ($x=1.02$) sample sintered at 1300°C for 2 h, maximized dielectric constant (ϵ_r) and $Q \times f$ value was 10.3 and 71,000 GHz, respectively. It is well known that the Q value was influenced by many factors including bulk density, grain boundary, secondary phase, and other defects (extrinsic) as well as anharmonic lattice vibration (intrinsic). It was believed that the variation in $Q \times f$ factor was primarily correlated with the bulk density in this system, except that CaMoO_4 (for $x=0.08$) and hot pressed sample, which would be influenced by another factor. For example, the $Q \times f$ value of hot pressed sample (55,300 GHz) was lower than that of pure CaMoO_4 sample (61,600 GHz), regardless of higher density.

4. Conclusion

In our study, pure CaMoO_4 sample shows lower sinterability. So as a way to obtain fully dense ceramics, the sample prepared by hot pressing (uniaxially pressed at 1100°C for 1 h under pressure of 16 MPa). Hot-pressed sample with $\sim 99\%$ of the theoretical density exhibited $Q \times f$ value of $\sim 55,000$ GHz and dielectric constant (ϵ_r) = 11.7. Because hot-press method was not suitable for commercializing, the samples with different Ca/Mo ratio were prepared to control sintering behavior of CaMoO_4 . Different Ca/Mo ratio system successfully improved the sintering and dielectric properties of CaMoO_4 , since MoO_3 rich liquid phase over the eutectic temperature act as sintering aid. $\text{CaMo}_{(x)}\text{O}_4$ (for $x=1.02$) sample sintered at 1300°C for 2 h with $\sim 95.7\%$ of the theoretical density shows $Q \times f$ value of 71,000 GHz and dielectric constant (ϵ_r) = 10.3.

Consequently, $\text{CaMo}_{(x)}\text{O}_4$ ceramic can be selected as a suitable candidate for microwave dielectric material because of its low dielectric constant and high- Q .

Acknowledgments

This research was supported by a grant from the Center for Advanced Materials Processing (CAMP) of the 21st Century Frontier R&D Program funded by the Ministry of Science and Technology, Republic of Korea.

References

- Johnson, L. F., Boyd, G. D., Nassau, K. and Soden, R. R., Continuous operation of a solid-state optical maser. *Phys. Rev.*, 1962, **126**(4), 1406–1409.
- Sharma, N., Shaju, K. M., Subba Rao, G. V., Chowdari, B. V. R., Dong, Z. L. and White, T. J., Carbon-coated nanophase CaMoO_4 as anode material for Li ion batteries. *Chem. Mater.*, 2004, **16**, 504–512.
- Kim, S. S., Ogura, S., Ikuta, H., Uchimoto, Y. and Wakihara, M., Reaction mechanisms of MnMoO_4 for high capacity anode material of Li secondary battery. *Solid State Ionics*, 2002, **146**, 249–256.
- Brower, W. S. and Fang, P. H., Dielectric constants of PbMoO_4 and CaMoO_4 . *Phys. Rev.*, 1966, **149**, 646.

5. Kaifez, D. and Guillion, P., *Dielectric Resonators*. Artech House, Norwood, MA, 1986, p. 327.
6. Hakki, B. W. and Coleman, P. D., Microwave theory and techniques. *IRE Trans.*, 1960, **8**, 402.
7. Nishikawa, T., Wakino, K., Tamura, H., Tanaka, H. and Ishikawa, Y., *IEEE Microwave Theory Tech. Symp. Dig.*, 1987, **3**, 277.
8. Yanushkevich, T. M. and Zhukovitskii, V. M., Phase diagram of the molybdenum trioxide-calcium oxide system. *Zh. Neorg. Khim.*, **18**(8), 2234–1973; *Russ. J. Inorg. Chem. (Engl. Transl.)*, 1973, **18**(8), 1182–1185.
9. Hanna, A. A. and Khilla, M. A., Electrical properties of the semiconductor materials molybdenum and tungsten trioxides. *Thermochim. Acta*, 1983, **65**, 311–320.